



1 Review

2 The role of zwitterionic materials in the fight against 3 proteins and bacteria

4 Montserrat Colilla^{1,2}, Isabel Izquierdo-Barba^{1,2} and María Vallet-Regí^{1,2,*}5 ¹ Departamento de Química en Ciencias Farmacéuticas. Facultad de Farmacia, Universidad Complutense de
6 Madrid. Instituto de Investigación Sanitaria Hospital 12 de Octubre i+12. Plaza Ramón y Cajal s/n, 28040
7 Madrid, Spain8 ² Center on Bioengineering, Biomaterials and Nanomedicine (CIBER-BBN), 28040 Madrid, Spain.9 * Correspondence: vallet@ucm.es; Tel.: +34-91-3941861

10 Received: date; Accepted: date; Published: date

11 **Abstract:** Zwitterionization of biomaterials has been heightened to a potent tool to develop
12 biocompatible materials able to inhibit bacterial and non-specific proteins adhesion. This
13 constitutes a major progress in the biomedical field. This manuscript overviews the main
14 functionalization strategies that have been reported up to date to design and develop these
15 advanced biomaterials. On this regard, the recent research efforts dedicated to provide their
16 surface of zwitterionic nature are summarized by classifying biomaterials in two main groups. First,
17 we centre on biomaterials in clinical use, concretely bioceramics and metallic implants. Finally, we
18 revise emerging nanostructured biomaterials, which are receiving growing attention due to their
19 multifunctionality and versatility mainly in the local drug delivery and bone tissue regeneration
20 scenarios.

21 **Keywords:** Biomaterials; zwitterionic surfaces; infection; bioceramics; bacterial inhibition.
22

23 1. Introduction

24 The increased usage of implantable medical devices is likely to result in a rise in the number of
25 infections associated to these cases. Bacterial contamination during biomaterial implantation is often
26 inevitable, provoking a battle between host cells and bacteria that may eventually cause the
27 infection. This is a devastating complication which presents a heterogeneous clinical profile being
28 considered as the most difficult infection disease to treat with serious clinical and socio-economic
29 implications [1,2]. Biomaterials-associated infections generally include bacterial adhesion,
30 colonization and biofilm formation on the biomaterial surfaces. In general, these infections are
31 mainly caused by different pathogens as *Staphylococcus epidermidis* and *Staphylococcus aureus* [3],
32 although *Escherichia coli* and *Pseudomonas aeruginosa* are also present [4]. These bacterial
33 colonizations result in an inflammatory reaction and are accompanied by significant morbidity and
34 mortality rate. For all these reasons, there is an urgent need to develop biomaterials with improved
35 properties able to provide solution to this serious clinical complication [4]. In this sense, preventive
36 strategies aimed at inhibiting the first stages of any infective process constitute a powerful and
37 promising alternative to tackle this issue [5].

38 Much research effort is being dedicated to develop new approaches to modify biomaterial
39 surface to inhibit the bacterial adhesion. This could be an attractive alternative to antibiotics, which
40 are associated with severe side effects, that moreover could create bacterial resistance. In this sense,
41 it has recently been recognized that both nanotopography and chemical surfaces show an essential
42 role in bacterial adhesion and biofilm formation [6]. In this sense, nanostructured surfaces currently
43 represent a good alternative as bacterial-repelling surfaces [7,8]. These surfaces comprise a variety of
44 nanotubes- or nanoparticle-based surfaces, and nanostructured coatings [9], which create a

45 superhydrophobic surface (also called leaves lotus effect) repelling the bacteria adhesion and
46 compromising in most of the cases to the host cell-tissue integration [10].

47 Concerning chemical modification to create bacterial-repelling surfaces, *zwitterionization* has
48 emerged as a revolutionary approach to provide biomaterials of high resistance to nonspecific
49 protein adsorption, bacterial adhesion and/or biofilm formation [11]. This strategy also allows the
50 preservation of the biomaterial biocompatibility in terms of host cell adhesion and colonization,
51 cytotoxicity and differentiation.

52 The aim of this review manuscript is to describe the different strategies developed so far for the
53 *zwitterionization* of biomaterials. First, we focus on biomaterials in clinical use, concretely
54 bioceramics and metallic implants. Later on, we revise the recent advances on nanostructured
55 biomaterials, which have gained much attention by the scientific community owing to their great
56 potential and versatility in local drug delivery and bone tissue regeneration. To understand the
57 processes underlying the different behaviors of *zwitterionic* surfaces by repelling bacteria meanwhile
58 allowing host cell colonization, an overview of the significance of the "race for the surface" concept
59 between bacteria and eukaryotic cells is also given.

60 2. Tuning the surface properties of biomaterials

61 The biocompatibility of the biomaterials is tightly related to the performance of cells when they
62 become in contact and adhere to its surface. In this regard, the surface features of these materials,
63 such as their topography chemistry or surface energy, become essential pillars in their
64 biocompatibility [6]. Moreover, depending of the bioceramic functionality, the specific requirements
65 in terms of cells/proteins adhesion are totally different. Therefore, in bone tissue regeneration
66 applications tuning the biomaterials surface to trigger bone bonding at the same time that inhibiting
67 bacterial colonization constitutes an exciting challenge to achieve better clinical outcomes.

68 Bacteria exist in nature under two states: planktonic (free-floating bacteria) and biofilm (sessile
69 microorganism communities). From all bacteria, just a small fraction ($\approx 1\%$) exists in the free floating
70 form, while around the 99% appear forming biofilms [12]. It is important to denote that the primary
71 and harder barrier in treating *S. aureus* infections is often attributed to the formation of bacterial
72 biofilm. In this sense, the biofilms are mainly built by big sessile bacteria communities embedded in
73 a self-produced extracellular polymeric matrix or glycocalyx [13]. Within the biofilm, bacteria grow
74 protected from environmental stress and resist attack by antibiotics, disinfectants, and the immune
75 system. [14]. Biofilm formation involves a sequence of four steps including: i) *adhesion*, initial
76 attachment of the bacteria to the surface of the tissue implant; ii) *growth*, bacterial aggregation and
77 accumulation in multiple bacterial layers; iii) *biofilm maturation* and iv) *dispersion*, detachment of
78 bacteria from the biofilm and spreading to other places (asepsis state) [15].

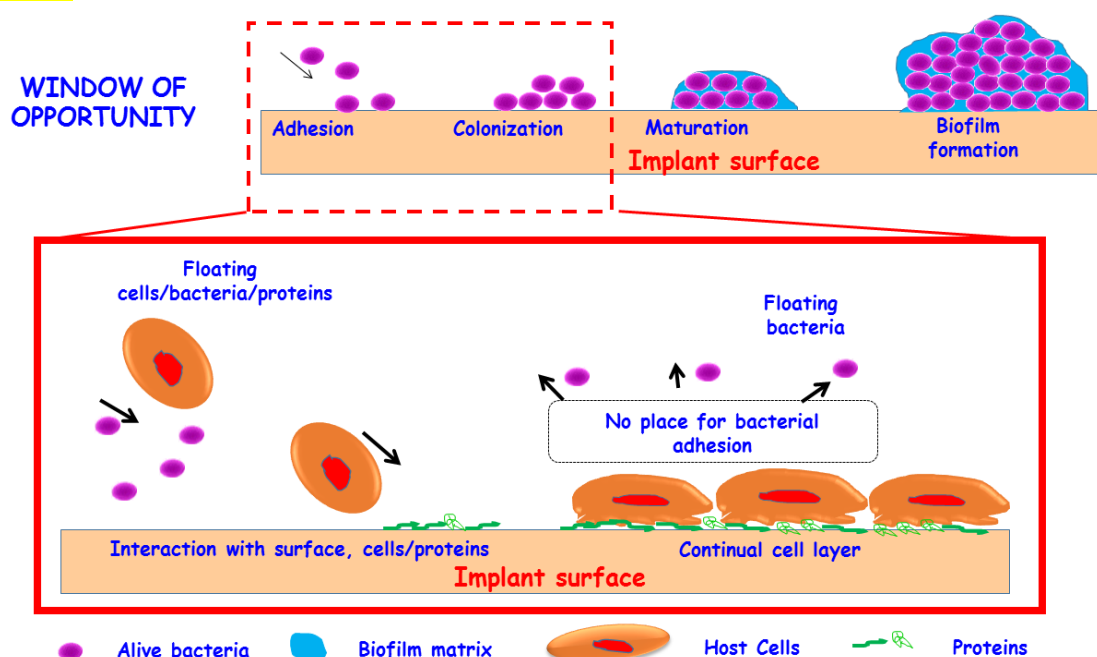
79 In 1987, the orthopedic surgeon Anthony G. Gristina described the concept of "race for the
80 surface" to predict the evolution of an implant in the specific relation to an infection process [16].
81 This concept contemplates "a race" between the eukaryotic cells (host cells) and the bacteria towards
82 the implant surface, arguing that when the host cells colonize its surface, the probability of bacterial
83 colonization is very low. However, when the implant is first colonized by the bacteria, it is
84 irreversibly infected, not allowing the eukaryotic cell to colonize it. Thus, such a concept has
85 stimulated technological and biomaterial progress while emphasizing the role of implant
86 biocompatibility and tissue-integration. Thus, the great challenge is to design implants that make the
87 race being won by eukaryotic cells, covering all biomaterial surfaces at the same time that inhibit
88 bacterial colonization. However, the "race for the surface" concept has been criticized for its
89 simplicity (simple rules) and the static conditions in which it is assessed, being useful notably for
90 specific purposes as determining surface affinity to different species and the effects of coexistence
91 [17]. Concerning this issue, the he most destabilizing factor is the great survival rate of the bacteria
92 since they are able to adhere and survive on any surface [18,19].

93 The bacterial adhesion process can be divided into two main phases as reversible and
94 irreversible stages, being the first phase mechanically and biologically less stable than the second
95 one [20,21]. The first phase, encompassing the bacteria adhesion and micro-colonies formation, is

96 mainly governed by the electrostatic attraction forces between bacteria and surfaces (mediated by
 97 certain proteins like adhesins) and where the electrochemical nature of the biomaterial plays a major
 98 role [22,23]. On the contrary, the second phase is governed by molecular and cellular interactions
 99 closely related with expression of specific gene clusters of the biofilm. They initiate the secretion of
 100 an protective slime formed by mucopolysaccharide layer, which becomes extremely resistant to both
 101 host immune system and antibiotic diffusion [24]. It is important to remark that bacteria cannot
 102 initiate the biofilm-related phenotype before they firmly attach to the implant. Thus, the transition
 103 phase between reversible and irreversible processes of biofilm formation constitutes the last
 104 “window of opportunity” for clinically reasonable preventive treatments.

105 On the other hand, concerning the host site, the way eukaryotic cells interact with an implant
 106 surface is through an interface that consists of discrete attachment protein points (integrins). These
 107 integrins interact with specific moieties of the extracellular matrix, such as RGD motifs [25], which
 108 contribute to bone regeneration and remodeling processes, being protected against bacterial
 109 colonization. However, neither osseointegration nor fibrous tissue encapsulation of an implant can
 110 eliminate long-term survival of bacterial micro-colonies, which also contribute to a possible delayed
 111 infection. As a result, there is a strong need to design an intrinsic implant with antibacterial
 112 functionality that can overcome and kill these remnant bacteria [26].

113 Figure 1 shows the process of biomaterial surface colonization by bacteria starting from a
 114 reversible phase where individual floating microorganisms settle down by low stable
 115 adhesin-mediated non-specific interactions with the biomaterial. These first steps delimit a “window
 116 of opportunity” for almost all antibiofilm strategies, in which is possible to inhibit the final biofilm
 117 formation and to reverse the final destiny of biomaterial for cell colonization. In this sense, if the host
 118 cell win the “race for the surface”, which attain irreversible attachments on the biomaterial surface
 119 first, the presence of a continual cell layer makes it complicated for bacteria adhesion and biofilm
 120 formation.



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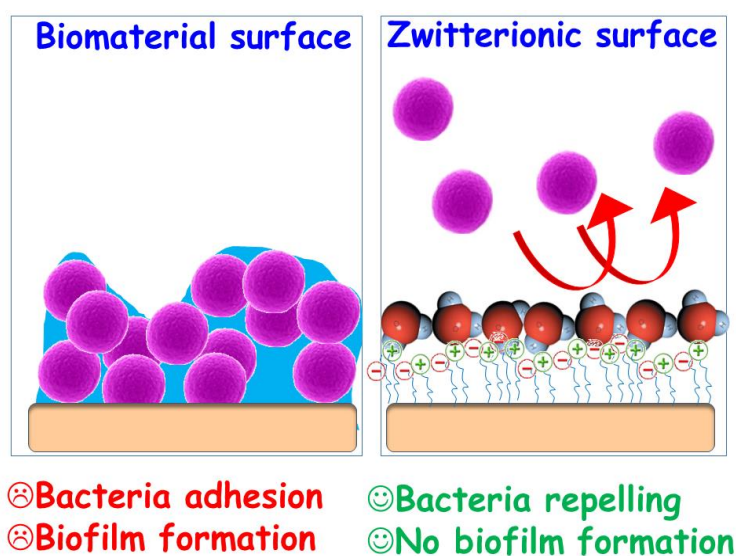
● Alive bacteria ● Biofilm matrix ● Host Cells ● Proteins

122 **Figure 1.** Concept of “race for the surface” and significance of “window of opportunity” in the
 123 development of implants capable to be colonized by host cells, while impede the bacteria adhesion
 124 and formation of the biofilm.

125 3. Significance of zwitterionization of biomaterials

126 The development of zwitterionic bioceramics to inhibit unspecific protein adsorption was
 127 reported for the first time in 2010 by Colilla *et al.* [27]. Previously, Jiang and co-workers had
 128 described the zwitterionization to develop polymeric and metallic biomaterials fulfilling the

129 ultralow-fouling criterion ($<5 \text{ ng/cm}^2$) [28,29]. *Zwitterionic* surfaces possess an equal number of both
 130 negatively and positively charged groups maintaining overall electrical neutrality, which depends
 131 on the pH of the environment [30]. Their non-fouling properties, as in the case of hydrophilic
 132 materials, are associated to the formation of a hydration layer on surface of biomaterial, forming a
 133 physical and energetic barrier that hinders unspecific proteins adhesion. Recently, zwitterionization
 134 of bioceramics has emerged as a cutting-edge technology to confer surfaces not only of high
 135 resistance to non-specific protein adsorption, but also to bacterial adhesion and/or biofilm formation
 136 (Figure 2) [11,31]. However the main requisite of any biomaterial, biocompatibility, must be kept in
 137 mind, i.e. zwitterionization must prevent bacterial adhesion but allow adequate host cell
 138 colonization. In vitro studies using osteoblastic-like cells revealed that they are able to appropriately
 139 adhere, colonize, and spread onto the surface of these zwitterionic bioceramics [31]. This different
 140 behaviour between prokaryotic and eukaryotic cells are caused by two main reasons [32]: i) the wall
 141 of the bacteria is formed by phospholipids layer, as eukaryotic cells, being much more rigid due to
 142 an external layer of peptidoglycan, ii) bacteria are much smaller in size (ca. $1 \mu\text{m}$) than eukaryotic
 143 cells (ca. $50 \mu\text{m}$). It has been proven that these both bacteria characteristics could be responsible of
 144 their capacity to discriminate differences in the biomaterials surfaces at the nanoscale level. As it has
 145 been above discussed in section 2, since eukaryotic cells adhere to surfaces via integrins-mediated
 146 mechanisms, bacteria adhesion is mainly driven by electrostatic attractive forces mediated by
 147 adhesions. In this last case, the electrochemistry of the biomaterial surface is an essential factor
 148 governing their adhesion [22,23], which explains the different behaviour compared to eukaryotic
 149 cells.



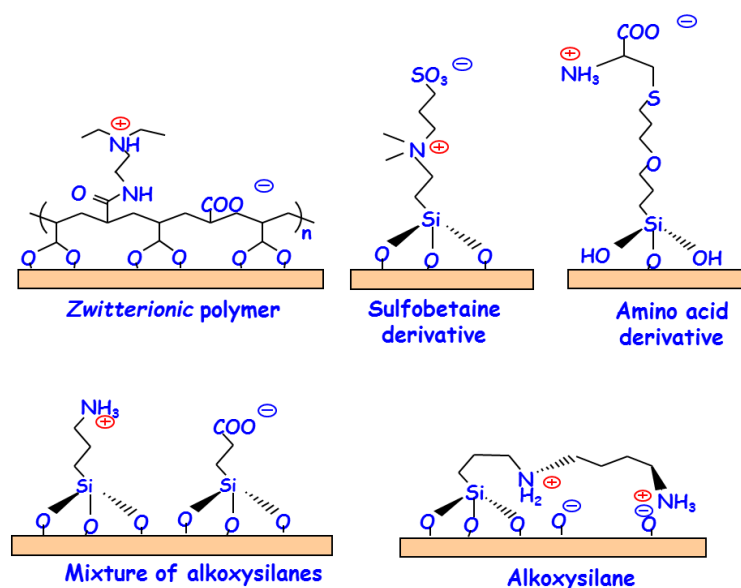
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151 **Figure 2.** Schematic illustration of the different behavior of conventional biomaterial surfaces vs
 152 biomaterials *zwitterionic* surfaces against bacterial colonization. As bacteria get close to the surface of
 153 conventional biomaterials, they are able to adhere, colonize and forming a biofilm, which is one of
 154 the major concerns in biomaterials associated infections. Oppositely, *zwitterionic* surfaces provide
 155 biomaterials of bacterial-repelling properties, thus inhibiting the subsequent biofilm formation,
 156 which constitutes a promising alternative in the biomaterials scenario to prevent bacterial infection.

157 3.1. Chemical strategies for the zwitterionization of biomaterials

158 In general terms, the *zwitterionization* of biomaterials involve in the functionalization of their
 159 surfaces at atomic level [11] In the beginning, the research efforts were focused on functionalizing
 160 with zwitterionic polymers bearing mixed positively and negatively charged moieties within the
 161 same chain and overall charge neutrality (Figure 3) [28,29,33-35]. There are three main
 162 methodologies to graft these polymers to the surface of biomaterials: (i) for functionalizing with
 163 poly(sulfobetaine) and polycarboxybetaine derivatives by surface-initiated atom transfer radical
 164 polymerization (SI-ATRP) [36-39]; (ii) for grafting sulfobetaine copolymers by surface reversible

165 addition-fragmentation chain transfer (RAFT) through the polymerization method denoted as
 166 graft-from-surface" and (iii) more simple procedures through polymerization method
 167 "graft-to-surface"[40]. In this method, polymers carrying adhesive moieties with strong surface
 168 affinity are synthesized and then grafted onto the surface through their adhesive moieties [41,42].
 169 Furthermore, it is possible to confer biomaterials of zwitterionic nature by decorating their surface
 170 with low-molecular weight moieties bearing the same number of negative and positive charges
 171 (Figure 3). For instance, it is possible to functionalize with different amino acids such as cysteine and
 172 lysine [43-45], sulfobetaine derivatives [45-47] or dopamine [48,50], exhibiting zwitterionic
 173 characteristics depending on the pH. Although the reported methods usually requires several
 174 synthetic steps involving different intermediate products, they offer distinct advantages compared
 175 to zwitterionic polymers, since they are usually associated to relatively more simple methods and
 176 lead to more biocompatible surfaces. A significant advance in the design and development of
 177 zwitterionic surfaces has consisted on the use of more direct and simple grafting methods by
 178 functionalization with organosilanes. In this case, the presence of hydroxyl (-OH) on the biomaterial
 179 surface helps to the simultaneously attach two organosilanes bearing positive and negative charges,
 180 respectively. This strategy allows for tailoring the zwitterionic properties by adjusting the molar
 181 ratio of the two organosilanes used during the synthesis. This process can be accomplished using
 182 two different alternatives, the co-condensation and the post-synthesis route. In the case of
 183 co-condensation method, functionalization takes place at the same time that the biomaterial is being
 184 synthesized. For instance, zwitterionic mesoporous SBA-15 material containing $-NH_3^+/-COO^-$
 185 groups was synthesized by adding two alkoxy silanes, aminopropyltrimethoxysilane (APTES) and
 186 carboxyethyl silanetriol sodium salt (CES) (Figure 3), together with the tetraethylorthosilicate
 187 (TEOS), as silica precursor, during the synthesis step [27]. Moreover, our research group has also
 188 reported the synthesis of zwitterionic SBA-15 by functionalization with
 189 (N-(2-aminoethyl)-3-aminopropyl-trimethoxysilane) (DAMO) alkoxy silane, which contains
 190 primary and secondary amine groups [52]. As it can be observed in Figure 3 its zwitterionic nature is
 191 provided by the presence of $-NH_3^+/-SiO^-$ and $>NH_2^+/-SiO^-$ zwitterionic pairs. On the other hand, the
 192 post-synthesis route relies on grafting the organosilanes to the biomaterials surface once they have
 193 been synthesized. Following this methodology, zwitterionic hydroxyapatite has been also been
 194 prepared by linking APTES and CES to the P-OH groups present in the surface of this biomaterial
 195 [53].



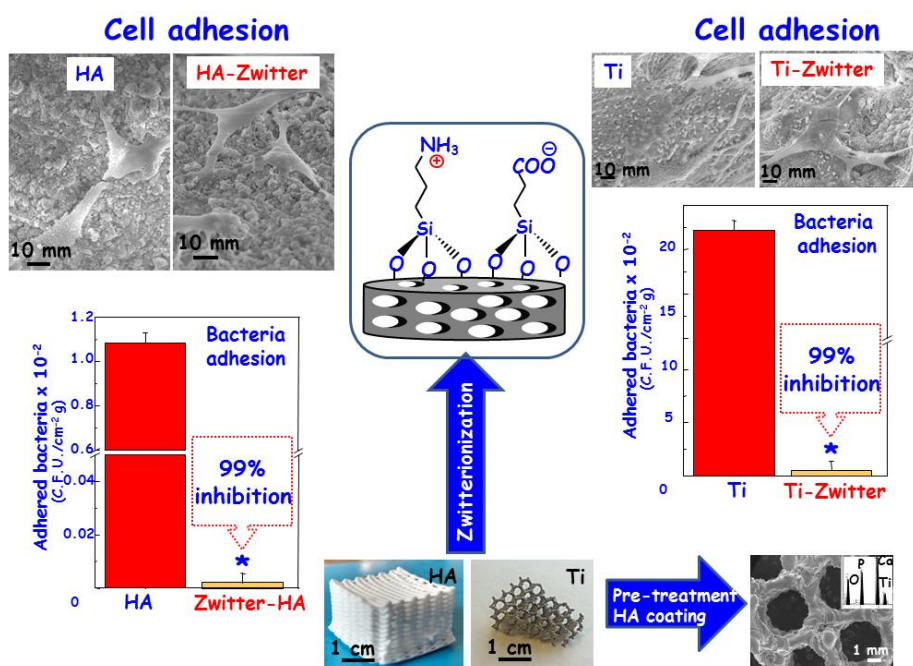
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197 **Figure 3.** Representative examples of the different chemical strategies developed so far for the
 198 zwitterionization of biomaterials: Grafting of zwitterionic polymers [e.g. 3-(diethylamino)propylamine
 199 (DEAPA) coupled to poly(acrylic acid) (PAA)]; Sulfobetaine siloxane derivatives; Amino acid
 200 derivatives (e.g. cysteine); Mixture of alkoxy silanes [e.g. 3-aminopropyltrimethoxysilane (APTES)

201 and carboxyethylsilanetriol sodium salt (CES); Alkoxysilane
202 [N-(2-aminoethyl)-3-aminopropyl-trimethoxysilane) (DAMO)].

203 3.2. Zwitterionization of biomaterials to prevent bacterial infection

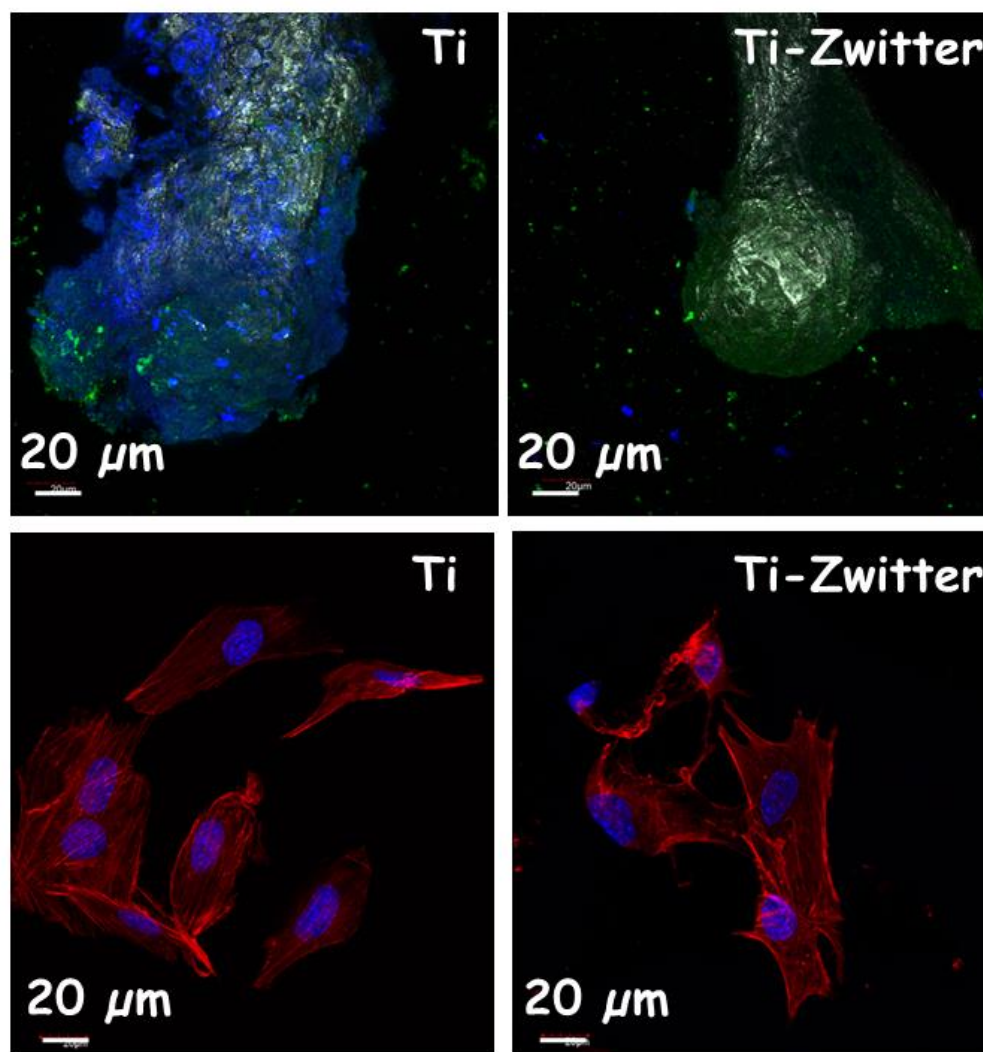
204 Currently, one of the major clinical challenges regarding the use of biomaterials is their
205 custom-made design depending on the biomedical application [54,55]. Bioceramic implants for bone
206 tissue regeneration, such as those based on calcium phosphates or bioactive glasses, can be
207 manufactured as three dimensional (3D) scaffolds using rapid prototyping (RP) methods [54]. These
208 scaffolds exhibit a high percentage of porosity and interconnectivity, and ease to be modulated with
209 improved mechanical properties compared to scaffolds fabricated by conventional methods [56].
210 Moreover, it is possible to combine nanostructural characteristics with micro-macro architecture for
211 a fine-tuning of cellular behavior [57-58]. However, when facing the regeneration of large and
212 critical bone defects, bioceramic implants are not suitable due to their intrinsic brittles [54]. It is
213 feasible to manufacture metallic alloys using these RP methods, providing strong scaffolding to the
214 bone regenerations purposes with porosities higher than 50% in volume, the rest being engaged by a
215 metal skeleton [59,60]. In this regard, the milestone in bone tissue regeneration is to design these 3D
216 scaffolds with surfaces capable of inhibiting and/or abolishing bacterial infection meanwhile
217 allowing osteoblast cells colonization. This surface would constitutes a great technological advance
218 to achieve better clinical outcomes. Currently, the scientific community is focussed for the design of
219 3D scaffolds that dynamically contribute to the regeneration process, stimulating the
220 osteoconduction and angiogenesis at the same time than evade the bacterial infection [60-63].
221 However, the challenge is to provide 3D scaffolds of zwitterionic character. In this sense, 3D
222 scaffolds based on pure nanocrystalline HA have been successfully constructed with a zwitterionic
223 nature by post-synthesis grafting of APTES and CES [53], which incorporates both $-NH_3^+$ and $-COO^-$
224 groups on the surface, respectively (Figure 4, Left). To attain this goal, HA 3D scaffolds were first
225 prepared by using RP technique and then the resulting 3D-HA scaffolds were bifunctionalized by
226 grafting both alkoxysilanes. Microbiological assays regarding bacterial adhesion using Escherichia
227 coli (E. coli) showed a noticeable inhibition of 99% with respect to unmodified 3D-HA. The
228 coexistence of $-NH_3^+/-COO^-$ pairs onto 3D-HA scaffold avails its bacterial-repelling properties. At
229 the same time, in vitro assays using HOS osteoblastic-like cells cultures demonstrated excellent
230 biocompatibility as the cells were able to spread and colonize the entire scaffold surface. Scanning
231 Electron Microscopy (SEM) micrographs show viable osteoblastic-like cells, exhibiting polygonal
232 shapes with filopodia-like projections attached to the surfaces (Figure 4, Left). Moreover, the cell
233 migration within the overall 3D-HA structure was demonstrated, showing total colonization of 3D
234 scaffold at different levels. This zwitterionization approach has been also applied onto Ti6Al4V
235 3D-scaffolds fabricated also by RP techniques [64]. In this case, to improve the functionalization
236 capability of the metallic surface, it was previously coated by a HA layer using the dip-coating
237 method. Once the HA coating was formed onto the Ti6Al4V 3D-scaffold, its surface was
238 zwitterionized by direct grafting of APTES and CES, following the same procedure as that reported
239 for pure 3D-HA scaffolds [53]. Again, the presence of zwitterionic pairs inhibits S. aureus adhesion
240 and biofilm formation, while permitting the osseointegration of this metallic implant, showing
241 MC3T3-E1 preosteoblasts colonizing the entire scaffold surface. The obtained results indicate that
242 the zwitterionization process does not affect the biocompatible properties of the metallic Ti6Al4V
243 3D-scaffolds, showing neither noticeable differences regarding cytotoxicity nor less proliferation
244 compared to bare 3D scaffold (Figure 4, right). Regarding the biofilm formation capability of these
245 surfaces, we carried out confocal microscopy to study the biofilm formation after 24 h of incubation
246 with S. aureus by using directly simultaneously acridine orange (green) and calcofluor (blue)
247 fluorescent dyes, which label live bacteria and extracellular matrix of biofilms, respectively (Figure
248 4, Top). The obtained results clearly display the biofilm formation by the blue staining of a typical
249 extracellular matrix covering the bacterial colonies with a thickness of $15 \pm 3.3 \mu m$ on the Ti6Al4V
250 scaffolds, while blue staining is absent in Ti-Zwitter scaffolds, revealing the non-formation of the
251 biofilm after 24 h of assay.



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253 **Figure 4.** Zwitterionization of different biomaterials in clinical use by grafting of
 254 carboxyethylsilanetriol sodium salt (CES) and aminopropyltriethoxysilane (APTES) onto the surface
 255 of these biomaterials. Left: Pure HA 3D Rapid Prototyping (RP) scaffolds. Right: Electron Beam
 256 Melting (EBM) Ti6Al4V 3D scaffolds.

257 Concerning the regeneration process, we carried out a simple assay by confocal microscopy
 258 using preosteoblast MC3T3-E1 seeded on the surface of these scaffolds and incubated during 7 days.
 259 Both colonization and cell-morphology were studied by staining the cytoskeleton with Atto
 260 565-conjugated phalloidin (red) and nuclei with DAPI (blue) as it can be observed in Figure 5
 261 (bottom). **In both cases**, the cells **display** a high spreading **grade** with a **well-built actin**
 262 cytoskeleton and high level of colonization in all entire surface of both scaffolds. These results revealed that the
 263 zwitterionization process does not affect the healing process, showing the same behavior that
 264 unmodified Ti [64]. In the case of zwitterionic biomaterials currently under research, nanostructured
 265 bioceramics are receiving growing attention by the scientific community. Among these
 266 nanostructured biomaterials, **silica-based mesoporous bioceramics** are in the crest of the wave
 267 because of their exceptional **features**, such as high surface areas and pore volumes, tunable and
 268 narrow pore size distributions and easy-to-functionalize surfaces [65-67]. Therefore they become
 269 excellent candidates to be provided of zwitterionic nature but also to host a great variety of
 270 antibiotics, allowing the combination of bacterial repellent and killing capabilities [68-71].



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Figure 5. Confocal microscopy images showing of EBM-Ti6Al4V 3D scaffolds before (Ti) and after being zwitterionized (Ti-Zwitter). *In vitro* behavior of the biomaterials after being incubated during 24 hours in *S. aureus* bacteria (Top) and during 4 days in MC3T3-E1 preosteoblast cells (Bottom). The results reveal the biofilm formation on the Ti surface, appearing the blue coating corresponding to the polysaccharide matrix (calcofluor), whereas this coating is not observed on the surface of Ti-zwitter surface, which confirms the antibiofilm formation preventing capability of this material. Both Ti and Ti-zwitter surfaces undergo an appropriate preosteoblastic colonization and spreading in the entire surface, with the nuclei stained in blue (DAPI) and the cytoskeleton stained in red (phalloidin).

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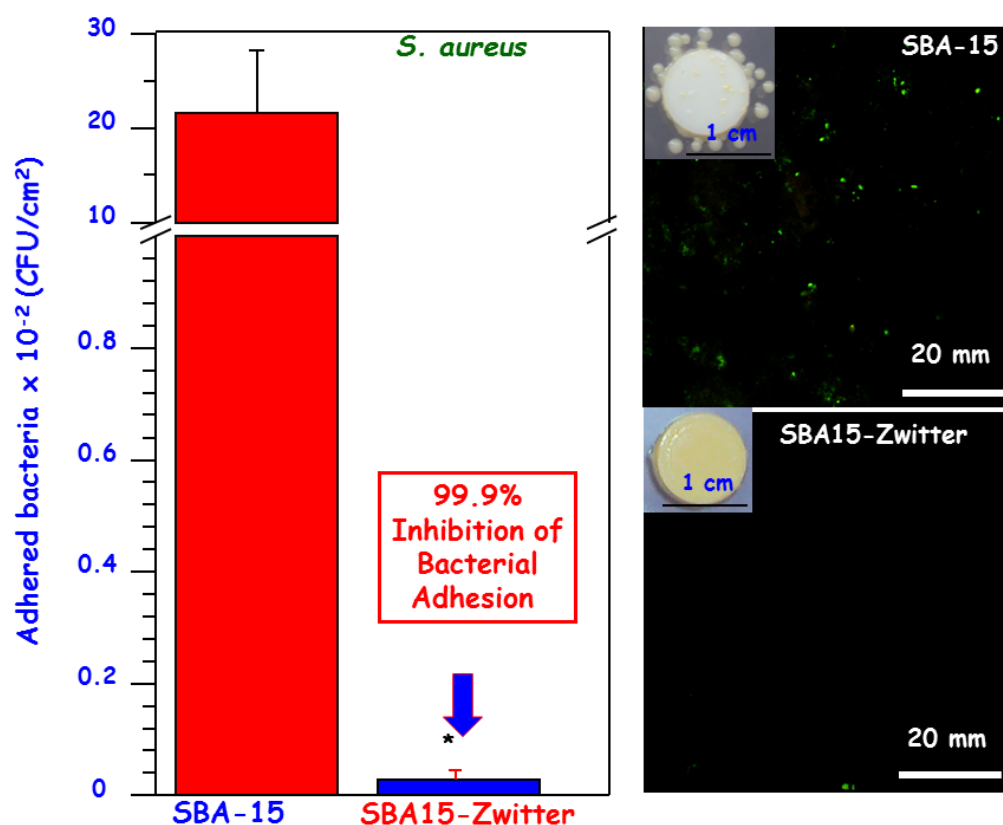
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Figure 6 shows the performance of zwitterionic SBA-15 (SBA15-Zwitter) nanostructured bioceramic owning $-\text{NH}_3^+/-\text{SiO}^-$ and $>\text{NH}_2^+/-\text{SiO}^-$ pairs [52], provided by the co-condensation functionalization with the diamine alkoxy silane (DAMO) (Figure 3). This bioceramic exhibits zwitterionic character at the physiological pH of 7.4, which constitutes a significant advance in this kind of materials for biomedical applications. *In vitro* bacterial adhesion tests using *S. aureus* strains reveal a great reduction of 99.9% with respect to unmodified SBA-15 (Figure 6). The intrinsic features of this nanostructured bioceramic **permits loading of the broad-spectrum antibiotic**, cephalixin, showing a loading capability of around $13 \text{ mg}\cdot\text{g}^{-1}$ together with a sustained drug release during more than 15 days. The synergistic combination of zwitterionic nature and antibiotic hosting capability opens up a new insight in the management of bone-associated infections. Zwitterionization has been also implemented on mesoporous bioactive glasses (MBG) having the $\text{SiO}_2\text{-CaO-P}_2\text{O}_5$ composition [44]. These MBG are a type are nanostructured bioceramics analogous in

293 composition to conventional bioglasses but exhibiting outstanding bioactive and cell response
 294 behaviors [69,72]. Thus, Sánchez-Salcedo et al. have recently reported the zwitterionization of MBG
 295 by tethering lysine (MBG-Lys). In vitro bacterial adhesion assays with *S. aureus* proved a reduction
 296 up to 99.9% compared to unmodified MBG. Moreover, MBG-Lys are cytocompatible, as
 297 demonstrated by *in vitro* studies carried out with MC3T3-E1 preosteoblasts cultures, which increases
 298 the potential application of these nanostructured bioceramics in bone tissue regeneration.



299

300 **Figure 6.** Schematic depiction of pure silica SBA-15 and *zwitterionic* SBA-15 (SBA-Zwitter)
 301 nanostructured materials. SBA-Zwitter was prepared following the co-condensation route in the
 302 presence of DAMO. Counting of colony forming units of *S. aureus* after 90 min of culture onto SBA-15
 303 and SBA15-Zwitter surfaces. Statistical significance: *p <0.01. Confocal microscopy images of *S.*
 304 *aureus* adhered onto SBA-15 and SBA15-Zwitter surfaces after staining with Baclight® Kit™.

305 4. Conclusions

306 *Zwitterionization* is emerging as a powerful strategy to design advanced biomaterials for the
 307 management of infection that is envisioned to result in better clinical outcomes. The possibility to
 308 easily functionalize biomaterials surface at the atomic and nanoscale levels to prevent the
 309 non-specific protein and bacterial adhesion opens up many paths to tackle severe clinical concerns.
 310 Indeed, *zwitterionic* biomaterials have shown an opposite behaviour by inhibiting bacterial adhesion
 311 while allowing host cells adhesion and colonization of the surface. This fact constitutes the
 312 cornerstone in their potential clinical application, and much research effort is committed to translate
 313 these significant advances from bench to bedside. However, there are certain challenges that these
 314 biomaterials have to face for their clinical stage of development. These challenges include: (i)
 315 preservation of biocompatibility, (ii) adequate pharmacokinetics through the local administration of
 316 antimicrobial agents and (iii) bone healing capacity, which guarantee success in bone regeneration.

317 **Author Contributions:** All the authors have participated equally in the elaboration of this review manuscript.

318 **Funding:** This research was funded by European Research Council through ERC-2015-AdG-694160
 319 (VERDI) grant.

320 **Acknowledgments:** The authors acknowledge financial support from European Research Council
321 through ERC-2015-AdG-694160 (VERDI) project.

322 **Conflicts of Interest:** The authors declare no conflict of interest.

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